

## Neodymium(III) molybdenum(VI) borate, $\text{NdBO}_2\text{MoO}_4$

Peter Held,\* Benjamin van der Wolf, Ladislav Bohatý and  
Petra Becker

Institut für Kristallographie, Universität zu Köln, Greinstraße 6, D-50939 Köln,  
Germany  
Correspondence e-mail: peter.held@uni-koeln.de

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Key indicators: single-crystal X-ray study;  $T = 292$  K; mean  $\sigma(\text{O}-\text{B}) = 0.006$  Å;  
 $R$  factor = 0.025;  $wR$  factor = 0.063; data-to-parameter ratio = 16.2.

Single crystals of  $\text{NdBO}_2\text{MoO}_4$  were obtained from a molybdenum oxide–boron oxide flux under an air atmosphere. The structure features double chains of edge- and face-sharing distorted  $[\text{NdO}_{10}]$  bicapped square-antiprisms, which are linked by rows of isolated  $[\text{MoO}_4]$  tetrahedra and by zigzag chains of corner-sharing  $[\text{BO}_3]$  groups, all of them running along the  $b$  axis. The chains of  $[\text{NdO}_{10}]$ , chains of  $[\text{BO}_3]$  and rows of  $[\text{MoO}_4]$  groups are arranged in layers parallel to the  $bc$  plane.

## Related literature

A rough investigation of subsolidus phase relations in the pseudo-ternary system  $\text{Nd}_2\text{O}_3 - \text{B}_2\text{O}_3 - \text{MoO}_3$  has been reported by Lysanova *et al.* (1983) and Dzhurinskii & Lysanova (1998). X-ray powder diffraction data of  $\text{LnBO}_2\text{MoO}_4$  ( $\text{Ln} = \text{La}, \text{Ce}, \text{Pr}, \text{Nd}$ ) are given by Lysanova *et al.* (1983). The occurrence of a structural phase transition of  $\text{LaBO}_2\text{MoO}_4$  was reported by Becker *et al.* (2008). For determinations of related structures, see: Palkina *et al.* (1979); Zhao *et al.* (2008, 2009) for  $\text{LaBO}_2\text{MoO}_4$ ; Zhao *et al.* (2008) for  $\text{CeBO}_2\text{MoO}_4$ .

## Experimental

### Crystal data

$\text{NdMoBO}_6$   
 $M_r = 346.99$

Monoclinic,  $P2_1/c$   
 $a = 10.1218$  (19) Å

$b = 4.1420$  (5) Å  
 $c = 11.896$  (3) Å  
 $\beta = 116.897$  (14)°  
 $V = 444.78$  (16) Å<sup>3</sup>  
 $Z = 4$

Mo  $K\alpha$  radiation  
 $\mu = 14.30$  mm<sup>-1</sup>  
 $T = 292$  K  
 $0.30 \times 0.20 \times 0.15$  mm

### Data collection

Enraf–Nonius CAD-4/MACH3 diffractometer  
Absorption correction:  $\psi$  scan (*MolEN*; Fair, 1990)  
 $T_{\min} = 0.487$ ,  $T_{\max} = 0.998$   
5004 measured reflections

1344 independent reflections  
1268 reflections with  $I > 2\sigma(I)$   
 $R_{\text{int}} = 0.035$   
3 standard reflections every 100 reflections  
intensity decay: 2.1%

### Refinement

$R[F^2 > 2\sigma(F^2)] = 0.025$   
 $wR(F^2) = 0.063$   
 $S = 1.22$   
1344 reflections

83 parameters  
 $\Delta\rho_{\max} = 2.24$  e Å<sup>-3</sup>  
 $\Delta\rho_{\min} = -1.61$  e Å<sup>-3</sup>

Data collection: *MACH3* (Enraf–Nonius, 1993); cell refinement: *MACH3*; data reduction: *MolEN* (Fair, 1990); program(s) used to solve structure: *SHELXS97* (Sheldrick, 2008); program(s) used to refine structure: *SHELXL97* (Sheldrick, 2008); molecular graphics: *ATOMS* (Dowty, 2002); software used to prepare material for publication: *publCIF* (Westrip, 2010).

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Supplementary data and figures for this paper are available from the IUCr electronic archives (Reference: FI2107).

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## **supplementary materials**

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### Comment

In the family of lanthanide compounds  $Ln\text{BO}_2\text{MoO}_4$  the crystal structure of  $\text{LaBO}_2\text{MoO}_4$  was first described by Palkina *et al.* (1979), giving the polar space group  $P2_1$  and lattice constants  $a = 5.964$  (1) Å,  $b = 4.147$  (1) Å,  $c = 9.373$  (3) Å,  $\beta = 99.28$  (2)°. On the basis of X-ray powder diffraction data, Lysanova *et al.* (1983) claimed the isomorphism of the compounds  $Ln\text{BO}_2\text{MoO}_4$  with  $Ln = \text{La}, \text{Ce}, \text{Pr}, \text{Nd}$ . The structure of  $\text{LaBO}_2\text{MoO}_4$  was redetermined by Zhao *et al.* (2008), who reported a centrosymmetric structure with space group symmetry  $P2_1/c$  with lattice constants  $a = 10.2968$  (8) Å,  $b = 4.1636$  (3) Å,  $c = 23.8587$  (15) Å,  $\beta = 115.367$  (3)°. These results were corroborated by Zhao *et al.* (2009), who, however, attributed this crystal structure to a high-temperature modification of  $\text{LaBO}_2\text{MoO}_4$ . The occurrence of a structural phase transition of  $\text{LaBO}_2\text{MoO}_4$  was reported by Becker *et al.* (2008). Zhao *et al.* (2008) also presented the crystal structure of the related cerium compound,  $\text{CeBO}_2\text{MoO}_4$ , with space group  $P2_1/c$  and lattice constants  $a = 10.2404$  (15) Å,  $b = 4.1495$  (4) Å,  $c = 11.9286$  (14) Å,  $\beta = 116.100$  (9)°, and thus showed, that the presumed isomorphism of the lanthanum and the cerium compound is not correct. However, the crystal structures are closely related, with an unit cell of  $\text{LaBO}_2\text{MoO}_4$  which is doubled with respect to the unit cell of  $\text{CeBO}_2\text{MoO}_4$ . The result of the present study shows, that also  $\text{NdBO}_2\text{MoO}_4$  is not isomorphic to  $\text{LaBO}_2\text{MoO}_4$ , but is isomorphic to  $\text{CeBO}_2\text{MoO}_4$ .

The crystal structure of  $\text{NdBO}_2\text{MoO}_4$  consists of groups  $[\text{BO}_3]$  and  $[\text{MoO}_4]$  and tenfold coordinated neodymium atoms. The  $[\text{BO}_3]$  groups are connected *via* common oxygen ligands to zigzag chains  $[\text{B}_2\text{O}_4]_\infty$  along the  $b$  axis, with a periodicity of two  $[\text{BO}_3]$  groups. The planar  $[\text{BO}_3]$  groups are slightly distorted with O—B—O angles of 114.4 (4)°, 117.4 (4)° and 128.1 (4)°. They are linked by the common oxygen atom O1 with a bond angle B—O1—B of 125.6 (3)° (see Fig. 1). The O1 ligands are connected to two boron atoms and one Nd atom, each, while the O2 ligands of the  $[\text{BO}_3]$  groups are linked to three Nd atoms and one B, each. Mo atoms are tetrahedrally coordinated by the oxygen atoms O3, O4, O5 and O6 (see Fig. 1), with Mo—O distances ranging from 1.740 (4) Å (Mo—O3) to 1.816 (3) Å (Mo—O4) and angles O—Mo—O ranging from 96.3 (2)° (O4—Mo—O3) to 117.5 (2)° (O4—Mo—O6). The  $[\text{MoO}_4]$  tetrahedra are arranged in rows that run parallel the  $b$  axis. Within a single row the  $[\text{MoO}_4]$  tetrahedra are aligned with their Mo—O3 bonding directions pointing in the same direction approximately parallel to the  $b$  axis. Rows with opposite Mo—O3 bonding direction ( $+b$  and  $-b$ ) alternate along the  $c$  axis, as shown in Fig. 2. Within a row the distance of a Mo atom to the O3 ligand of the neighbouring tetrahedron amounts 2.419 (3) Å. (Note that in Zhao *et al.* (2008) a distorted trigonal bipyramidal is preferred as description of the coordination surrounding of Mo in  $\text{CeBO}_2\text{MoO}_4$ .) The oxygen atoms O3 and O5 serve as ligands of one Mo and one Nd atom, each, while the oxygen atoms O6 and O4 act as ligands for one Mo and two Nd atoms, each. The tenfold coordination of neodymium atoms in  $\text{NdBO}_2\text{MoO}_4$  can be described as distorted bicapped square antiprism with Nd—O bonding distances ranging between 2.364 (3) Å (Nd—O2) and 2.981 (4) Å (Nd—O6).  $[\text{NdO}_{10}]$  polyhedra are connected along the  $b$  axis by sharing three common oxygen ligands, namely O2, O6 and O4 (Fig. 1), with each neighbouring polyhedron, thus forming chains along the  $b$  axis. Two parallel chains are linked *via* edges (formed by two common O2 atoms) of the Nd coordination polyhedra, resulting in a double chain along  $b$ , see Fig. 2.

# supplementary materials

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## Experimental

$\text{NdBO}_2\text{MoO}_4$  melts incongruently, therefore, single crystals of  $\text{NdBO}_2\text{MoO}_4$  were grown from a melt with a composition differing from the crystal stoichiometry. A homogenized powder mixture of  $\text{Nd}_2\text{O}_3$  (99.9%, Alfa Aesar),  $\text{B}_2\text{O}_3$  (99.98%, Alfa Aesar) and  $\text{MoO}_3$  (99.95%, Alfa Aesar) in a molar ratio of 1: 1.375: 2.625 was heated in a covered platinum crucible in air atmosphere to 1423 K and subsequently cooled at a rate of 3 K h<sup>-1</sup> to 1173 K. Transparent violet prismatic crystals of the title compound were separated mechanically from the surrounding flux. A suitable single-crystal was carefully selected using a polarizing microscope and mounted in a glass capillary.

## Refinement

The final difference maps indicate a maximum of 2.235 e Å<sup>-3</sup> at a distance of 0.66 Å from Nd and a minimum of -1.611 e Å<sup>-3</sup> at a distance of 1.06 Å of Nd.

## Figures

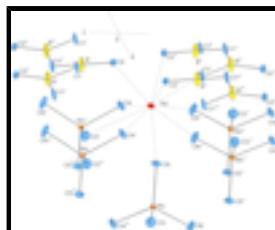


Fig. 1. The structural units in  $\text{NdBO}_2\text{MoO}_4$  with atomic numbering scheme (projection approximately along the  $b$ -axis). Atoms are drawn as 50% probability ellipsoids. Symmetry codes: (i)  $x, y + 1, z$ ; (ii)  $-x, -y + 1, -z - 1$ ; (iii)  $-x, -y, -z - 1$ ; (iv)  $-x + 1, -y, -z - 1$ ; (v)  $-x + 1, y - 1/2, -z - 1/2$ ; (vi)  $-x + 1, -y + 1, -z - 1$ ; (vii)  $-x + 1, y + 1/2, -z - 1/2$ ; (viii)  $x, y - 1, z$ ; (ix)  $-x, y - 1/2, -z - 1/2$ ; (x)  $-x, y + 1/2, -z - 1/2$ ; (xi)  $x, -y + 1/2, z + 1/2$ ; (xii)  $x, -y - 1/2, z - 1/2$ ; (xiii)  $x, 3/2 - y, z - 1/2$ ; (xiv)  $1 - x, 1/2 + y, z - 1/2$ .

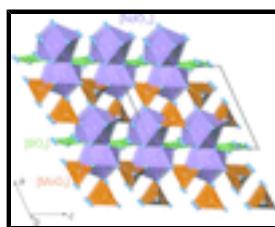


Fig. 2. View of the crystal structure of  $\text{NdBO}_2\text{MoO}_4$  along the  $b$  axis, showing chains of corner-sharing  $[\text{BO}_3]$  groups (green), double-chains of face- and edge-sharing  $[\text{NdO}_{10}]$  polyhedra (purple), and rows of isolated  $[\text{MoO}_4]$  tetrahedra (orange), all of them running along the  $b$  axis. In a single row,  $[\text{MoO}_4]$  tetrahedra are arranged with identical orientation, thus giving a polarity  $+b$  or  $-b$  to the row. Note the alternating polarity of the arrangement of  $[\text{MoO}_4]$  tetrahedra rows along the  $c$  axis.

## Neodymium(III) molybdenum(VI) borate

### Crystal data

$\text{NdMoBO}_6$	$F(000) = 620$
$M_r = 346.99$	$D_x = 5.182 \text{ Mg m}^{-3}$
Monoclinic, $P2_1/c$	Mo $K\alpha$ radiation, $\lambda = 0.71073 \text{ \AA}$
Hall symbol: -P 2ybc	Cell parameters from 25 reflections
$a = 10.1218 (19) \text{ \AA}$	$\theta = 12.1\text{--}21.2^\circ$
$b = 4.1420 (5) \text{ \AA}$	$\mu = 14.30 \text{ mm}^{-1}$
$c = 11.896 (3) \text{ \AA}$	$T = 292 \text{ K}$
$\beta = 116.897 (14)^\circ$	Prism, light violet
$V = 444.78 (16) \text{ \AA}^3$	$0.30 \times 0.20 \times 0.15 \text{ mm}$
$Z = 4$	

## *Data collection*

Enraf–Nonius CAD4/MACH3 diffractometer	1268 reflections with $I > 2\sigma(I)$
Radiation source: fine-focus sealed tube graphite	$R_{\text{int}} = 0.035$
$\omega/2\theta$ scans	$\theta_{\max} = 30.4^\circ$ , $\theta_{\min} = 2.3^\circ$
Absorption correction: $\psi$ scan ( <i>MolEN</i> ; Fair, 1990)	$h = -14 \rightarrow 14$
$T_{\min} = 0.487$ , $T_{\max} = 0.998$	$k = -5 \rightarrow 5$
5004 measured reflections	$l = -16 \rightarrow 16$
1344 independent reflections	3 standard reflections every 100 reflections intensity decay: 2.1%

## *Refinement*

Refinement on $F^2$	Primary atom site location: structure-invariant direct methods
Least-squares matrix: full	Secondary atom site location: difference Fourier map
$R[F^2 > 2\sigma(F^2)] = 0.025$	$w = 1/[\sigma^2(F_o^2) + (0.0304P)^2 + 1.8082P]$ where $P = (F_o^2 + 2F_c^2)/3$
$wR(F^2) = 0.063$	$(\Delta/\sigma)_{\max} < 0.001$
$S = 1.22$	$\Delta\rho_{\max} = 2.24 \text{ e } \text{\AA}^{-3}$
1344 reflections	$\Delta\rho_{\min} = -1.61 \text{ e } \text{\AA}^{-3}$
83 parameters	Extinction correction: <i>SHELXL97</i> (Sheldrick, 2008), $F_c^* = kFc[1 + 0.001xFc^2\lambda^3/\sin(2\theta)]^{1/4}$
0 restraints	Extinction coefficient: 0.0428 (13)

## *Special details*

**Geometry.** All e.s.d.'s (except the e.s.d. in the dihedral angle between two l.s. planes) are estimated using the full covariance matrix. The cell e.s.d.'s are taken into account individually in the estimation of e.s.d.'s in distances, angles and torsion angles; correlations between e.s.d.'s in cell parameters are only used when they are defined by crystal symmetry. An approximate (isotropic) treatment of cell e.s.d.'s is used for estimating e.s.d.'s involving l.s. planes.

**Refinement.** Refinement of  $F^2$  is done against ALL reflections. The weighted  $R$ -factor  $wR$  and goodness of fit  $S$  are based on  $F^2$ , conventional  $R$ -factors  $R$  are based on  $F$ , with  $F$  set to zero for negative  $F^2$ . The threshold expression of  $F^2 > 2\sigma(F^2)$  is used only for calculating  $R$ -factors(gt) etc. and is not relevant to the choice of reflections for refinement.  $R$ -factors based on  $F^2$  are statistically about twice as large as those based on  $F$ , and  $R$ -factors based on ALL data will be even larger.

## *Fractional atomic coordinates and isotropic or equivalent isotropic displacement parameters ( $\text{\AA}^2$ )*

	$x$	$y$	$z$	$U_{\text{iso}}^*/U_{\text{eq}}$
Nd	0.19204 (2)	0.21132 (6)	-0.47183 (2)	0.00919 (11)
Mo	0.64536 (4)	0.30466 (8)	-0.31742 (3)	0.00668 (12)
B	-0.0028 (7)	0.3290 (12)	-0.3083 (5)	0.0182 (10)
O1	-0.0326 (4)	0.6580 (8)	-0.3013 (3)	0.0120 (6)
O2	-0.0035 (3)	0.2247 (7)	-0.4150 (3)	0.0099 (5)

## supplementary materials

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O3	0.6616 (4)	0.7223 (8)	-0.3016 (4)	0.0173 (7)
O4	0.2605 (4)	-0.2856 (7)	-0.3501 (3)	0.0124 (6)
O5	0.4556 (4)	0.2256 (9)	-0.3956 (3)	0.0156 (6)
O6	0.7424 (4)	0.2194 (9)	-0.4084 (4)	0.0190 (7)

### Atomic displacement parameters ( $\text{\AA}^2$ )

	$U^{11}$	$U^{22}$	$U^{33}$	$U^{12}$	$U^{13}$	$U^{23}$
Nd	0.00733 (14)	0.00900 (16)	0.00992 (14)	-0.00094 (6)	0.00274 (10)	-0.00205 (7)
Mo	0.00602 (17)	0.0087 (2)	0.00546 (17)	-0.00022 (10)	0.00270 (13)	-0.00094 (10)
B	0.040 (3)	0.007 (2)	0.013 (2)	0.0020 (19)	0.017 (2)	-0.0003 (17)
O1	0.0259 (16)	0.0056 (13)	0.0050 (12)	-0.0010 (11)	0.0074 (12)	-0.0004 (10)
O2	0.0108 (13)	0.0118 (14)	0.0084 (13)	-0.0018 (10)	0.0055 (11)	-0.0021 (10)
O3	0.0203 (16)	0.0125 (15)	0.0199 (16)	0.0003 (12)	0.0099 (14)	-0.0002 (12)
O4	0.0173 (15)	0.0091 (14)	0.0067 (13)	-0.0007 (10)	0.0018 (12)	-0.0017 (10)
O5	0.0066 (12)	0.0235 (17)	0.0137 (15)	-0.0009 (11)	0.0019 (11)	-0.0008 (12)
O6	0.0271 (18)	0.0192 (17)	0.0207 (17)	-0.0024 (13)	0.0197 (15)	-0.0039 (13)

### Geometric parameters ( $\text{\AA}$ , $^\circ$ )

Nd—O2	2.364 (3)	B—O2	1.338 (6)
Nd—O5	2.399 (3)	B—O1 <sup>ix</sup>	1.381 (6)
Nd—O4	2.431 (3)	B—O1	1.406 (6)
Nd—O4 <sup>i</sup>	2.452 (3)	B—Nd <sup>ii</sup>	3.099 (6)
Nd—O1 <sup>ii</sup>	2.498 (3)	B—Nd <sup>iii</sup>	3.315 (6)
Nd—O2 <sup>iii</sup>	2.528 (3)	O1—B <sup>x</sup>	1.381 (6)
Nd—O6 <sup>iv</sup>	2.551 (3)	O1—Nd <sup>ii</sup>	2.498 (3)
Nd—O3 <sup>v</sup>	2.901 (4)	O2—Nd <sup>iii</sup>	2.528 (3)
Nd—O2 <sup>ii</sup>	2.930 (3)	O2—Nd <sup>ii</sup>	2.930 (3)
Nd—O6 <sup>vi</sup>	2.981 (4)	O3—Mo <sup>i</sup>	2.419 (3)
Mo—O3	1.740 (4)	O3—Nd <sup>vii</sup>	2.901 (4)
Mo—O5	1.745 (3)	O4—Mo <sup>v</sup>	1.816 (3)
Mo—O6	1.795 (3)	O4—Nd <sup>viii</sup>	2.452 (3)
Mo—O4 <sup>vii</sup>	1.816 (3)	O6—Nd <sup>iv</sup>	2.551 (3)
Mo—O3 <sup>viii</sup>	2.419 (3)	O6—Nd <sup>vi</sup>	2.981 (4)
Mo—Nd <sup>vii</sup>	3.5017 (8)		
O2—Nd—O5	145.40 (12)	O4 <sup>vii</sup> —Mo—Nd <sup>vi</sup>	128.02 (11)
O2—Nd—O4	84.21 (11)	O3 <sup>viii</sup> —Mo—Nd <sup>vi</sup>	120.27 (9)
O5—Nd—O4	80.04 (12)	Nd <sup>vii</sup> —Mo—Nd <sup>vi</sup>	103.052 (17)
O2—Nd—O4 <sup>i</sup>	81.98 (11)	Nd <sup>v</sup> —Mo—Nd <sup>vi</sup>	135.816 (15)
O5—Nd—O4 <sup>i</sup>	77.71 (12)	O3—Mo—Nd	99.98 (12)
O4—Nd—O4 <sup>i</sup>	116.06 (13)	O5—Mo—Nd	7.61 (12)
O2—Nd—O1 <sup>ii</sup>	95.08 (11)	O6—Mo—Nd	121.06 (13)
O5—Nd—O1 <sup>ii</sup>	117.89 (12)	O4 <sup>vii</sup> —Mo—Nd	113.32 (11)

O4—Nd—O1 <sup>ii</sup>	134.27 (11)	O3 <sup>viii</sup> —Mo—Nd	87.93 (9)
O4 <sup>i</sup> —Nd—O1 <sup>ii</sup>	109.02 (10)	Nd <sup>vii</sup> —Mo—Nd	114.650 (15)
O2—Nd—O2 <sup>iii</sup>	68.97 (12)	Nd <sup>v</sup> —Mo—Nd	105.813 (15)
O5—Nd—O2 <sup>iii</sup>	131.40 (11)	Nd <sup>vi</sup> —Mo—Nd	116.653 (16)
O4—Nd—O2 <sup>iii</sup>	69.84 (11)	O3—Mo—Nd <sup>iv</sup>	122.17 (12)
O4 <sup>i</sup> —Nd—O2 <sup>iii</sup>	149.91 (11)	O5—Mo—Nd <sup>iv</sup>	102.87 (12)
O1 <sup>ii</sup> —Nd—O2 <sup>iii</sup>	67.44 (11)	O6—Mo—Nd <sup>iv</sup>	20.25 (12)
O2—Nd—O6 <sup>iv</sup>	129.71 (11)	O4 <sup>vii</sup> —Mo—Nd <sup>iv</sup>	113.64 (11)
O5—Nd—O6 <sup>iv</sup>	72.70 (12)	O3 <sup>viii</sup> —Mo—Nd <sup>iv</sup>	60.20 (9)
O4—Nd—O6 <sup>iv</sup>	70.40 (12)	Nd <sup>vii</sup> —Mo—Nd <sup>iv</sup>	134.533 (16)
O4 <sup>i</sup> —Nd—O6 <sup>iv</sup>	148.09 (12)	Nd <sup>v</sup> —Mo—Nd <sup>iv</sup>	94.773 (16)
O1 <sup>ii</sup> —Nd—O6 <sup>iv</sup>	75.67 (12)	Nd <sup>vi</sup> —Mo—Nd <sup>iv</sup>	60.259 (12)
O2 <sup>iii</sup> —Nd—O6 <sup>iv</sup>	61.79 (10)	Nd—Mo—Nd <sup>iv</sup>	110.359 (15)
O2—Nd—O3 <sup>v</sup>	75.40 (11)	O2—B—O1 <sup>ix</sup>	128.1 (4)
O5—Nd—O3 <sup>v</sup>	70.07 (11)	O2—B—O1	117.4 (4)
O4—Nd—O3 <sup>v</sup>	58.78 (10)	O1 <sup>ix</sup> —B—Nd <sup>ii</sup>	155.4 (4)
O4 <sup>i</sup> —Nd—O3 <sup>v</sup>	57.31 (10)	O1 <sup>ix</sup> —B—Nd <sup>iii</sup>	102.0 (3)
O1 <sup>ii</sup> —Nd—O3 <sup>v</sup>	163.84 (10)	O1—B—Nd <sup>iii</sup>	129.6 (4)
O2 <sup>iii</sup> —Nd—O3 <sup>v</sup>	119.20 (10)	Nd <sup>ii</sup> —B—Nd <sup>iii</sup>	80.37 (14)
O6 <sup>iv</sup> —Nd—O3 <sup>v</sup>	120.47 (11)	O1 <sup>ix</sup> —B—Nd	120.0 (3)
O2—Nd—O2 <sup>ii</sup>	69.96 (11)	O1—B—Nd	111.8 (3)
O5—Nd—O2 <sup>ii</sup>	122.40 (10)	Nd <sup>ii</sup> —B—Nd	84.34 (13)
O4—Nd—O2 <sup>ii</sup>	154.13 (10)	Nd <sup>iii</sup> —B—Nd	74.17 (11)
O4 <sup>i</sup> —Nd—O2 <sup>ii</sup>	62.95 (10)	O2—B—Nd <sup>xi</sup>	144.1 (4)
O1 <sup>ii</sup> —Nd—O2 <sup>ii</sup>	50.40 (10)	O1—B—Nd <sup>xi</sup>	90.2 (3)
O2 <sup>iii</sup> —Nd—O2 <sup>ii</sup>	98.46 (10)	Nd <sup>ii</sup> —B—Nd <sup>xi</sup>	142.25 (16)
O6 <sup>iv</sup> —Nd—O2 <sup>ii</sup>	125.50 (11)	Nd <sup>iii</sup> —B—Nd <sup>xi</sup>	132.90 (16)
O3 <sup>v</sup> —Nd—O2 <sup>ii</sup>	113.50 (9)	Nd—B—Nd <sup>xi</sup>	117.93 (18)
O2—Nd—O6 <sup>vi</sup>	120.96 (10)	B <sup>x</sup> —O1—B	125.6 (3)
O5—Nd—O6 <sup>vi</sup>	73.22 (11)	B <sup>x</sup> —O1—Nd <sup>ii</sup>	132.3 (3)
O4—Nd—O6 <sup>vi</sup>	152.84 (11)	B—O1—Nd <sup>ii</sup>	101.4 (3)
O4 <sup>i</sup> —Nd—O6 <sup>vi</sup>	62.98 (10)	B <sup>x</sup> —O1—Nd <sup>xi</sup>	57.4 (2)
O1 <sup>ii</sup> —Nd—O6 <sup>vi</sup>	59.13 (11)	B—O1—Nd <sup>xi</sup>	68.4 (3)
O2 <sup>iii</sup> —Nd—O6 <sup>vi</sup>	126.02 (10)	Nd <sup>ii</sup> —O1—Nd <sup>xi</sup>	169.04 (12)
O6 <sup>iv</sup> —Nd—O6 <sup>vi</sup>	96.66 (11)	B <sup>x</sup> —O1—Nd	137.1 (3)
O3 <sup>v</sup> —Nd—O6 <sup>vi</sup>	114.35 (10)	B—O1—Nd	49.7 (3)
O2 <sup>ii</sup> —Nd—O6 <sup>vi</sup>	52.35 (9)	Nd <sup>ii</sup> —O1—Nd	78.20 (8)
O3—Mo—O5	105.76 (17)	Nd <sup>xi</sup> —O1—Nd	96.89 (8)
O3—Mo—O6	102.12 (17)	B—O2—Nd	129.3 (3)
O5—Mo—O6	114.36 (17)	B—O2—Nd <sup>iii</sup>	114.5 (3)
O3—Mo—O4 <sup>vii</sup>	96.26 (16)	Nd—O2—Nd <sup>iii</sup>	111.03 (12)

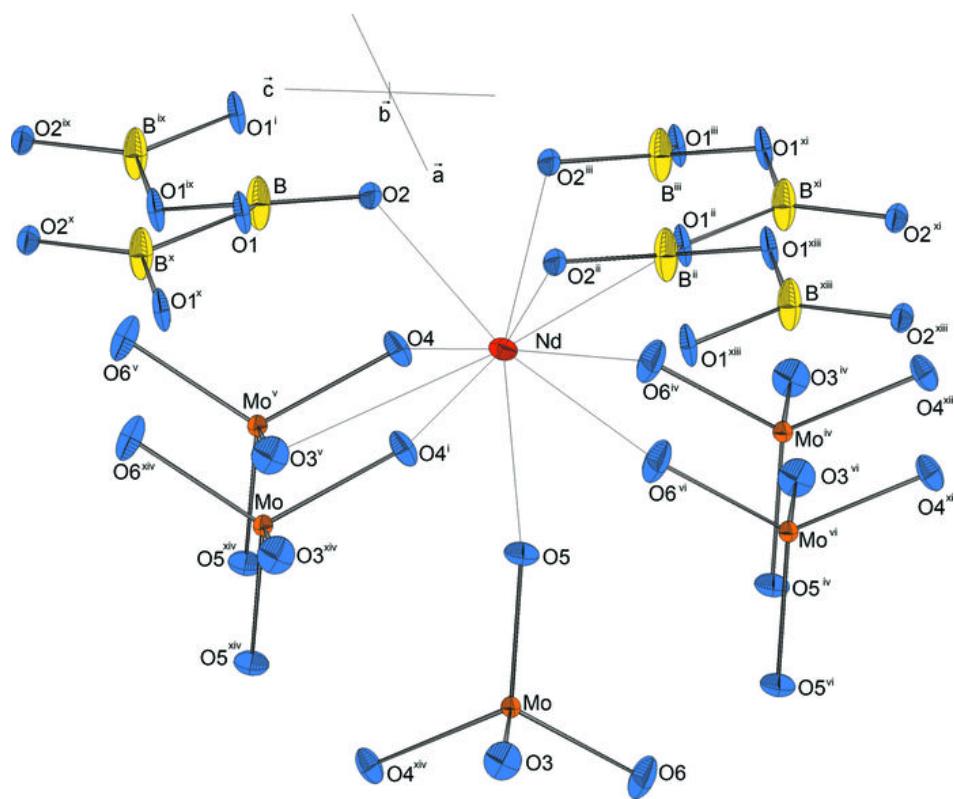
## supplementary materials

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O5—Mo—O4 <sup>vii</sup>	116.79 (16)	B—O2—Nd <sup>ii</sup>	84.3 (3)
O6—Mo—O4 <sup>vii</sup>	117.52 (17)	Nd—O2—Nd <sup>ii</sup>	110.04 (11)
O3—Mo—O3 <sup>viii</sup>	169.5 (2)	Nd <sup>iii</sup> —O2—Nd <sup>ii</sup>	98.46 (10)
O5—Mo—O3 <sup>viii</sup>	82.75 (15)	Mo—O3—Mo <sup>i</sup>	169.5 (2)
O6—Mo—O3 <sup>viii</sup>	79.30 (15)	Mo—O3—Nd <sup>vii</sup>	94.65 (15)
O4 <sup>vii</sup> —Mo—O3 <sup>viii</sup>	74.10 (13)	Mo <sup>i</sup> —O3—Nd <sup>vii</sup>	94.88 (12)
O3—Mo—Nd <sup>vii</sup>	55.67 (13)	Mo—O3—Nd <sup>vi</sup>	92.60 (14)
O5—Mo—Nd <sup>vii</sup>	121.91 (12)	Mo <sup>i</sup> —O3—Nd <sup>vi</sup>	84.42 (10)
O6—Mo—Nd <sup>vii</sup>	123.02 (13)	Nd <sup>vii</sup> —O3—Nd <sup>vi</sup>	131.41 (12)
O4 <sup>vii</sup> —Mo—Nd <sup>vii</sup>	40.63 (10)	Mo <sup>v</sup> —O4—Nd	110.25 (14)
O3 <sup>viii</sup> —Mo—Nd <sup>vii</sup>	114.71 (9)	Mo <sup>v</sup> —O4—Nd <sup>viii</sup>	133.69 (16)
O3—Mo—Nd <sup>v</sup>	123.03 (13)	Nd—O4—Nd <sup>viii</sup>	116.06 (13)
O5—Mo—Nd <sup>v</sup>	105.84 (12)	Mo <sup>v</sup> —O4—Nd <sup>iii</sup>	117.45 (14)
O6—Mo—Nd <sup>v</sup>	106.15 (13)	Nd—O4—Nd <sup>iii</sup>	71.19 (8)
O4 <sup>vii</sup> —Mo—Nd <sup>v</sup>	26.80 (10)	Nd <sup>viii</sup> —O4—Nd <sup>iii</sup>	78.89 (8)
O3 <sup>viii</sup> —Mo—Nd <sup>v</sup>	47.32 (9)	Mo—O5—Nd	166.9 (2)
Nd <sup>vii</sup> —Mo—Nd <sup>v</sup>	67.430 (16)	Mo—O6—Nd <sup>iv</sup>	145.65 (19)
O3—Mo—Nd <sup>vi</sup>	62.27 (12)	Mo—O6—Nd <sup>vi</sup>	115.69 (16)
O5—Mo—Nd <sup>vi</sup>	114.60 (12)	Nd <sup>iv</sup> —O6—Nd <sup>vi</sup>	96.66 (11)
O6—Mo—Nd <sup>vi</sup>	41.02 (12)		

Symmetry codes: (i)  $x, y+1, z$ ; (ii)  $-x, -y+1, -z-1$ ; (iii)  $-x, -y, -z-1$ ; (iv)  $-x+1, -y, -z-1$ ; (v)  $-x+1, y-1/2, -z-1/2$ ; (vi)  $-x+1, -y+1, -z-1$ ; (vii)  $-x+1, y+1/2, -z-1/2$ ; (viii)  $x, y-1, z$ ; (ix)  $-x, y-1/2, -z-1/2$ ; (x)  $-x, y+1/2, -z-1/2$ ; (xi)  $x, -y+1/2, z+1/2$ .

Fig. 1



## supplementary materials

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Fig. 2

